Mixed Solid Sorbents for CO₂ Capture Technologies: Theoretical Predictions and **Experimental Validations**

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I. Introduction

Carbon dioxide is one of the major combustion products which once released into the air can contribute to global climate change.¹⁻⁴ One approach to solve such environmental problems is to capture and sequester the CO₂. Current technologies for capturing CO₂ including solvent-based (amines) and CaO-based materials are still too energy intensive. Hence, there is critical need for development of new materials that can capture and release CO₂ reversibly with acceptable energy costs. In particular, solid oxide sorbent materials have been proposed for capturing CO₂ through a reversible chemical transformation leading primarily to formation of carbonate products. Solid sorbents containing alkali and alkaline earth metals have been reported in several previous studies to be promising candidates for CO₂ sorbent applications due to their high CO_2 absorption capacity at moderate working temperatures.⁵⁻⁸

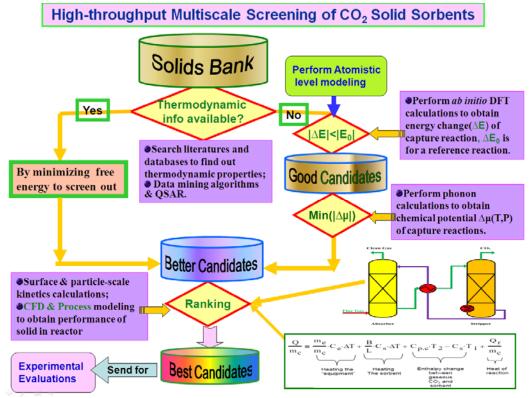


Fig. 1. Schematic of our screening methodology.

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During past few years we developed a theoretical methodology to identify promising solid sorbent candidates for CO₂ capture by combining thermodynamic database searching with *ab initio* thermodynamics obtained based on first-principles density functional theory (DFT) and lattice phonon dynamics.^{5-7, 9-13} As shown in Fig.1, the primary outcome of our screening scheme is a list of promising CO₂ sorbents with optimal energy usage.

At a given CO₂ pressure, the turnover temperature (T_t) of an individual solid capture CO₂ reaction is fixed. Such T_t may be outside the operating temperature range (ΔT_o) for a particularly capture technology. In order to adjust T_t to fit the practical ΔT_o , in this study, we demonstrate that by mixing different types of solids it's possible to shift T_t to the range of practical operating conditions.

II. Calculation Methods for Mixed Solid Sorbents

The complete description of the computational methodology together with relevant applications can be found in our previous publications.^{5-7, 9-16} The CO_2 capture reactions of solids can be expressed generically in the form (for convenient description, we normalized the reaction to 1 mole of CO_2)

$$\sum_{R_i} n_{R_i} \text{Solid} \ R_i + CO_2 \leftrightarrow \sum_{P_j} n_{P_j} \text{Solid} \ P_j$$
(a)

where n_{Ri} , n_{Pj} are the numbers of moles of reactants (R_i) and products (P_j) involved in the capture reactions. We treat the gas phase CO_2 as an ideal gas. By assuming that the difference between the chemical potentials ($\Delta \mu^{o}$) of the solid phases of reactants (R_i) and products (P_j) can be approximated by the difference in their total energies (ΔE^{DFT}), obtained directly from DFT calculations, and the vibrational free energy of the phonons and by ignoring the PV contribution terms for solids, the variation of the chemical potential ($\Delta \mu$) for reaction (a) with temperature and pressure can be written as^{5-7, 9-16}

$$\Delta\mu(T,P) = \Delta\mu^{0}(T) - RT \ln \frac{P_{CO_{2}}}{P_{0}}$$
(1)

where

$$\Delta \mu^{0}(T) \approx \Delta E^{DFT} + \Delta E_{ZP} + \Delta F^{PH}(T) - G^{0}_{CO_{2}}(T)$$
(2)

Here, ΔE^{DFT} is the DFT energy difference between the reactants and products of the reaction (a), ΔE_{ZP} is the zero point energy difference between the reactants and products and can be obtained directly from phonon calculations. ΔF^{PH} is the phonon free energy change excluding zero-point energy (which is already counted into the ΔE_{ZP} term) between the solids of products and reactants. P_{CO_2} is the partial pressure of CO_2 in the gas phase and P_0 is the standard state reference pressure taken to be 1 bar. The heat of reaction ($\Delta H^{cal}(T)$) can be evaluated through the following equation

$$\Delta H^{cal}(T) = \Delta \mu^{0}(T) + T[\Delta S_{PH}(T) - S_{CO_{2}}(T)]$$
(3)

where $\Delta S_{PH}(T)$ is the difference of entropies between product solids and reactant solids. The free energy of CO_2 ($G^0_{CO_2}$) can be obtained from standard statistical mechanics,^{6, 7, 13} and its entropy (S_{CO_2}) can be found in the empirical thermodynamic databases.¹⁷

III. Results and Discussions

When we mix two solids A and B to form a new sorbent C, the turnover temperature of the newly resulted system (T_c) is located between those of A and B (T_A , T_B). Here it was assumed that A is a strong

CO₂ sorbent while *B* is a weak CO₂ sorbent and $T_A > T_B$. Also, we assumed that the desired operating temperature T_O is between T_A and T_B ($T_A > T_O > T_B$). Now, depending on the properties of *A* and *B*, we have typically three scenarios to synthesize the mixing sorbent *C*:

3.1. $T_A >> T_B$ and the *A* component is the key part to capture CO₂.

An example of this case is represented by Li₂O. This is a very strong CO₂ sorbent which forms Li₂CO₃. However, its regeneration from Li₂CO₃ only can occur at very high temperature (T_A). In order to move its T_A to lower temperatures, one can mix some weak CO₂ sorbents (such as SiO₂, ZrO₂). Fig. 2 shows the relationship of chemical potential, P_{CO2}, and T of the CO₂ capture reactions by the mixed Li₂O/SiO₂ and Li₂O/ZrO₂ solids with different mixing ratios. Fig. 3 shows the turnover T and the CO₂ capture capacity of Li₂O/SiO₂ mixture versus the ratio of Li₂O/SiO₂.^{5-7, 9, 15, 16, 18, 19}

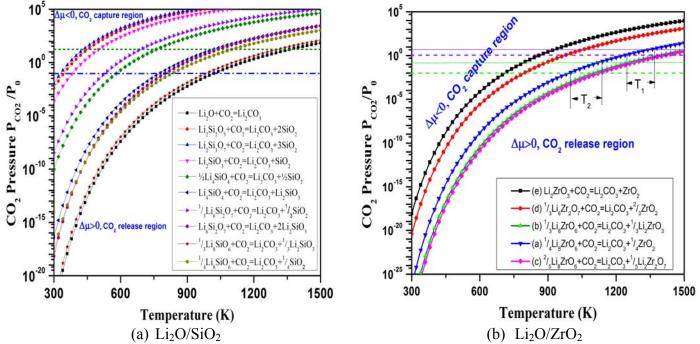


Fig. 2. Contour plots of the calculated chemical potential ($\Delta\mu$) versus temperature and the CO₂ pressure (*P* plotted in logarithmic scale) for the CO₂ capture reactions. Only $\Delta\mu$ =0 curve is shown explicitly. For each reaction, above its $\Delta\mu$ =0 curve, their $\Delta\mu$ <0, which means the sorbents absorb CO₂ and the reaction goes forward, whereas below the $\Delta\mu$ =0 curve, their $\Delta\mu$ >0, which indicates CO₂ start to be released and reaction goes backward with regeneration of the sorbents.

From Figs. 2 and 3, one can see that after mixing Li_2O and SiO_2 (or ZrO_2) with different Li_2O/SiO_2 (or Li_2O/ZrO_2) ratios, the T_C of the newly formed *C* compound (silicate or zirconate) is lower than T_A of pure Li_2O and could be close to the ΔT_o range to fit the practical needs.

From Figs. 2 and 3, one can see that during absorption of CO₂, the lithium silicates with Li₂O/SiO₂ ratio>1 (*e.g.* Li₈SiO₆, Li₆Si₂O₇, Li₄SiO₄) can be fully converted into SiO₂ and Li₂CO₃ because thermodynamically Li₂SiO₃ can further absorb CO₂ to gain energetic advantage. As shown in Fig. 12, for these Li₂O-rich lithium silicates, in order to obtain maximum CO₂ capture capacity, the operating temperatures during capture should be lower than the turnover temperatures (T₁ for pre-combustion capture or T₂ for post-combustion capture) of Li₂SiO₃. If the capture temperature is higher than that temperature for Li₂SiO₃, the reverse reaction (Li₂CO₃+SiO₂=Li₂SiO₃+CO₂) will start and the products will be Li₂SiO₃ and

 Li_2CO_3 . In this case, the CO_2 capture capacities of these Li_2O -rich lithium silicates could not reach their maxima. However, during regeneration, when Li_2CO_3 and Li_2SiO_3 (or SiO_2) react with each other to release CO_2 and regenerate the sorbent, the temperature is the key factor to ensure that the original lithium silicate being regenerated. Actually, from Fig. 12 one can see that from the "CO₂ capture region" to the "CO₂ release region" we can have different regenerating temperatures for different silicates.

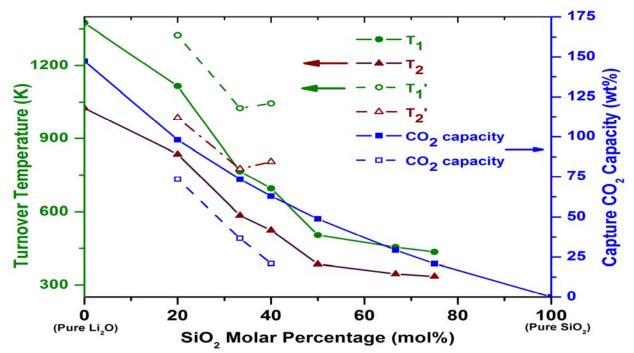


Fig. 3 The dependence of the turnover temperatures defined in the text and of CO₂ capture capacity on molar percentage of SiO₂ in the silicates for which calculations are reported here. T_1 and T_1 ' are the turnover temperatures under pre-combustion conditions with CO₂ partial pressure at 20 bars, while T_2 and T_2 ' are the turnover temperatures under post-combustion conditions with CO₂ partial pressure at 0.1 bar. The solid lines indicate to convert lithium silicates to SiO₂ and Li₂CO₃ (T_1 , T_2). For those Li₂O-rich lithium silicates (Li₈SiO₆, Li₄SiO₄, Li₆Si₂O₇) capturing CO₂, the data shown in dash lines indicate convert them to Li₂SiO₃ and Li₂CO₃ (T_1 ', T_2 '). The corresponding CO₂ capture capacities are plotted with open blue squares.

Among these lithium silicates, Fig. 4 shows our dynamic TGA data on the CO₂ capture of these lithium silicates, in addition to the Li₂O. From these curves, it is clearly seen how the Li₂O/SiO₂ ratio modified the amount of CO₂ captured and the temperature range in which the process is performed. We note that these experiments are not quantitative, thus the weight increase analysis must be performed in other types of experiments presented below. From the temperature point of view, it is evident that all the silicates captured CO₂ in two different steps, as it was previously described. Analyzing the Li₂O, as example, the first CO₂ capture process occurred between 180 and 380 °C. The weight increase at this stage is associated with the CO₂ superficial reaction. Then, once the diffusion processes are activated, the second weight increment was produced between 580 and 710 °C. Here, the CO₂ capture is produced in the silicate bulk. A similar interpretation is possible for the other curves depicted as inset in Fig. 4, although the reaction process and the external shell composition may differ in each lithium silicate.

The maximum temperature, at which each lithium silicate is able to trap CO_2 , has the following trend: $Li_2O > Li_8SiO_6 > Li_4SiO_4 > Li_2SiO_3$, at which the CO_2 capture properties of Li_2SiO_3 can be considered as

negligible. These results are in good agreement with our theoretical results presented in Figs. 2 and 3. Therefore, based on these results, if the non-tested lithium silicates were evaluated as possible CO_2 capture sorbent, the Li₆Si₂O₇ (Li₂O/SiO₂ ratio higher than that of Li₂SiO₃) could present some interesting properties.

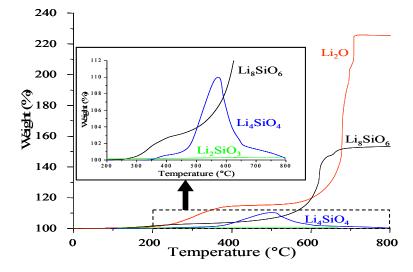


Fig.4. Dynamic thermogravimetric analyses of different lithium silicates (Li₂O, Li₈SiO₆, Li₄SiO₄ and Li₂SiO₃) into a CO₂ flux (60 mL/min).^{18, 19}

Fig. 5 shows the CO₂ isothermal weight gain due to CO₂ absorption as a function of time the same lithium silicates, at 600 °C. Qualitatively, at short times (~10min) the CO₂ capture rates for the lithium silicates are ordered as follows (fastest first): $Li_2O > Li_8SiO_6 > Li_4SiO_4 > Li_2SiO_3$, and the same trend is observed for long times (~80min). The only significant variation of this tendency was observed between Li_2O and Li_8SiO_6 : At large values of weight increase (long times) the order inverted and Li_8SiO_6 started to absorb faster than Li_2O . This change was observed at a specific time, where the diffusion process must begin to control the reaction process. Therefore, different microstructural features may have induced this change. In addition, although none of the isothermal curves reached equilibrium, Li_2O is the one which possesses the highest CO₂ absorption capacity at 600 °C for a long time.

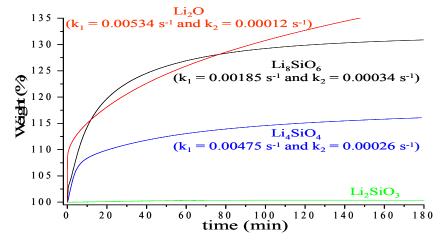


Fig.5. CO_2 isothermal experiments performed at 600 °C in the following lithium silicates Li_8SiO_6 , Li_4SiO_4 , Li_2SiO_3 and Li_2O . The k constants values reported correspond to the isothermal fitting to a double exponential model.

3.2 $T_A >> T_B$ and *B* component is the key part to capture CO₂

In this case, since T_B is lower than T_O , by mixing *A* into *B* will increase the turnover temperature T_C of the *C* solid to values closer to T_o . For example, pure MgO has a very high theoretical CO₂ capture capacity. However, its turnover temperature (250 °C) is lower than the required temperature range of 300-470 °C used in warm gas clean up technology and its practical CO₂ capacity is very low, and therefore, it cannot be used directly as a CO₂ sorbent in this technology.^{20, 21}

As shown in Fig. 6, by mixing alkali metal oxides M₂O (M=Na, K, Cs, Ca) or carbonates (M₂CO₃) into MgO, the corresponding newly formed mixing systems have higher turnover temperatures making them useful as CO₂ sorbents through the reaction MgO + CO₂ + M₂CO₃ = M₂Mg(CO₃)₂.^{20, 22}

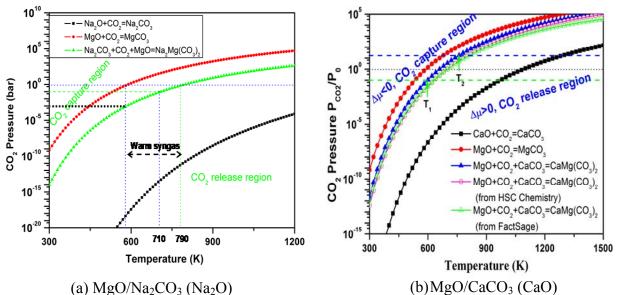


Fig. 6. Plots of the calculated chemical potentials versus CO₂ pressures and temperatures for the CO₂ capture reaction by MgO to form double salts.

Fig.7 shows the TSA and PSA test results comparison test of Na-promoted MgO sorbent. As one can see the CO₂ capture capacity is kept constant after first couple cycles.

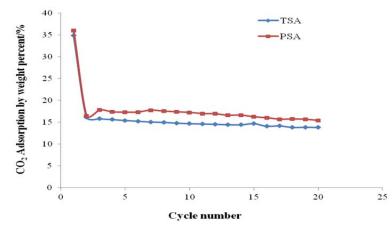


Fig.7. Na-promoted MgO sorbent 20-cycle TSA and PSA test results.

Fig.8 shows the x-ray diffraction pattern of Na-promoted MgO sorbent after loaded with CO₂. Clearly to see, there is double salt $Na_2Mg(CO_3)_2$ formed.^{20, 21}

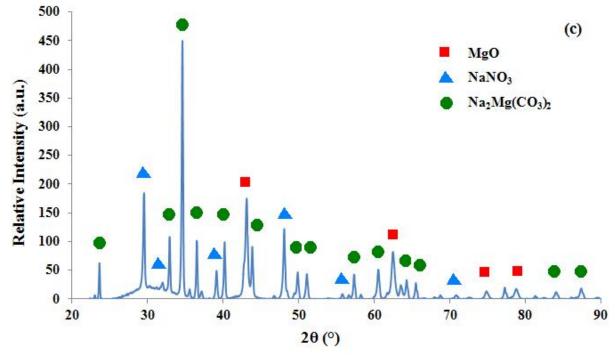


Fig.8. Na-promoted MgO sorbent after test, loaded with CO₂ x-ray diffraction patterns

Table 1. The weight percentage of CO₂ capture, the calculated energy change ΔE^{DFT} , the zero-point energy changes ΔE^{ZP} and the thermodynamic properties (ΔH , ΔG) of the CO₂ capture reactions. (unit: kJ/mol). The turnover temperatures (T₁ and T₂) of the reactions of CO₂ capture by solids under the conditions of pre-combustion (P_{CO2}=20 bar) and post-combustion (P_{CO2}=0.1 bar) are also listed.

	absorbing	ΔE^{DFT}	ΔE^{ZP}	ΔH	ΔG	Turnover	T (K)
reaction	$CO_2 Wt $	ΔE	ΔE	(T=300K)	(T=300K)	T ₁	T ₂
Li ₂ ZrO ₃ +CO ₂ ↔Li ₂ CO ₃ +ZrO ₂ ^a	28.75	-146.648	11.311	-158.562	-103.845	1000	780
LiNaZrO ₃ A+CO ₂ \leftrightarrow $\frac{1}{2}$ (Li ₂ CO ₃ +Na ₂ CO ₃)+ZrO ₂	26.01	-152.936	7.069	-176.666	-110.892	805	685
$LiNaZrO_3 B+CO_2 \leftrightarrow \frac{1}{2}(Li_2CO_3+Na_2CO_3)+ZrO_2$	26.01	-167.872	6.934	-191.526	-126.477	865	745
$LiKZrO_3 A+CO_2 \leftrightarrow \frac{1}{2}(Li_2CO_3+K_2CO_3)+ZrO_2$	23.75	-264.115	6.006	-287.513	-225.611	1275	1095
$LiKZrO_3 B+CO_2 \leftrightarrow \frac{1}{2}(Li_2CO_3+K_2CO_3)+ZrO_2$	23.75	-311.604	7.080	-332.612	-272.410	hT ^b	1285
Na ₂ ZrO ₃ +CO ₂ ↔Na ₂ CO ₃ +ZrO ₂ ^a	23.76	-140.862	2.236	-158.327	-114.121	1275	925
$K_2ZrO_3+CO_2\leftrightarrow K_2CO_3+ZrO_2^a$	20.24	-223.158	5.813	-238.490	-187.884	hT ^b	1285

^a from Refs.^{15, 18, 23}.

^b hT means the temperature is higher than our temperature range (1500K)

3.3 T_A and T_B are close to each other and both A and B are active to capture CO₂

In this case, both A and B components are active to capture CO_2 , and the CO_2 capacity of the mixture is the summation of those of A and B. As we know another potential advantage of mixing solids is to increase

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the surface area of the solids in order to have faster reaction rates. Such a mixing scenario doesn't show too much advantage in shifting the capture temperature, but may enhance the kinetics of the capture process and eventually make the mixtures more efficient. Although there is no such report in literature, we think such an attempt is worthwhile and are working on several doped systems.

Currently, we are working on doping systems. Here we presented the results of LiMZrO₃ (M=Na, K) capturing CO₂. In this case, we substituted half of Li in Li₂ZrO₃ with Na or K. The calculated results are shown in Figure 9 and Table 1.²⁴ Obviously, such doping system also can be treated as mixing three oxides, Li₂O, ZrO₂, M₂O, M=Na, K. The doping level/mixing ratios could change their thermodynamic properties to fit the industrial needs.

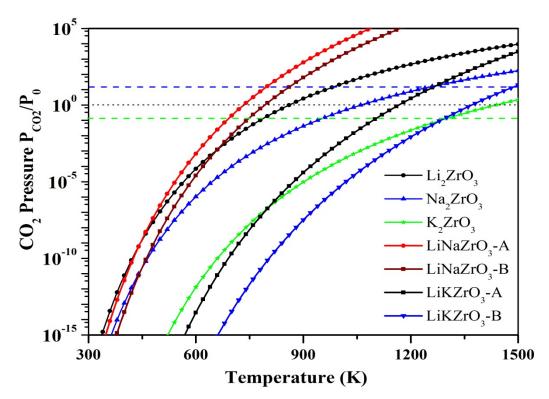


Fig. 9. The contour plotting of calculated chemical potentials versus CO₂ pressures and temperatures of the CO₂ capture reactions by LiMZrO₃ and M₂ZrO₃. Y-axis plotted in logarithm scale. Only the $\Delta\mu$ =0 curve is shown explicitly. For each reaction, above its $\Delta\mu$ =0 curve, their $\Delta\mu$ <0, which means the solids absorb CO₂ and the reaction goes forward, whereas below the $\Delta\mu$ =0 curve, their $\Delta\mu$ >0, which means the CO₂ start to release and the reaction goes backward to regenerate the sorbents.

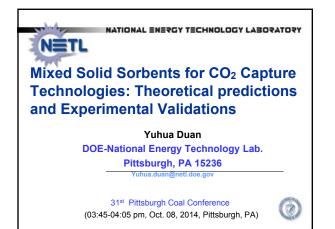
As one can see from Fig. 9 and Table 1, our results showed that by doping Na into Li_2ZrO_3 , the obtained new solid LiNaZrO₃ has better performance as a CO₂ sorbent for post-combustion capture technologies. For K doping into Li_2ZrO_3 , our calculated thermodynamic results showed that the new solid LiKZrO₃ does not gain improvement on its CO₂ capture performance because its regeneration temperature is much higher than Li_2ZrO_3 . Further detailed analyzes with different ratios of doping/mixing are under the way.

IV. Conclusions

The obtained results showed that by changing the mixing ratio of solid *A* and solid *B* to form mixed solid *C* it's possible to shift the turnover T_t of the newly formed solid *C* to fit the practical CO₂ capture technologies. When mixing SiO₂ or ZrO₂ into the strong Li₂O sorbent, one can obtain a series of lithium silicates (or zirconates) with T_t lower than that of pure Li₂O. By mixing oxides (Na₂O, K₂O, CaO) or their corresponding carbonates into MgO, the obtained mixtures exhibit different thermodynamic behaviors and their T_t are higher than that of pure MgO. Such results can be used to provide insights for designing new CO₂ sorbents. Therefore, although one single material taken in isolation might not be an optimal CO₂ sorbent to fit the particular needs to operate at specific temperature and pressure conditions, by mixing or doping two or more materials to form a new material, our results showed that it is possible to synthesize new CO₂ sorbent formulations which can fit the industrial needs. Our results also show that computational modeling can play a decisive role for identifying materials with optimal performance.

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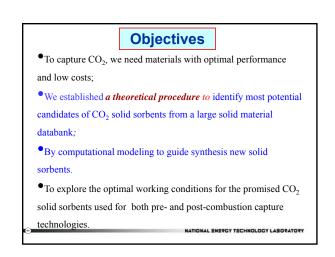
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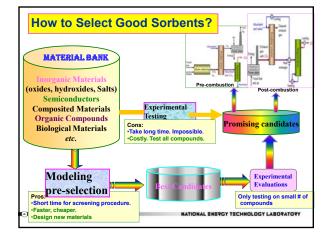
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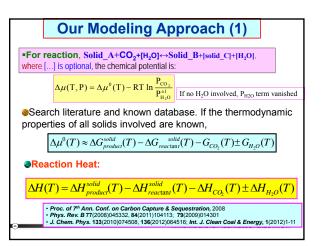
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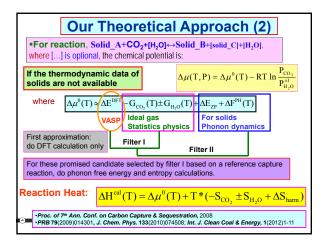
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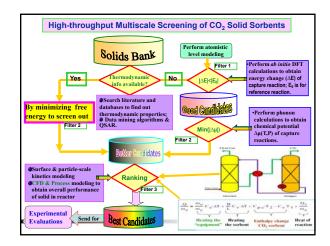
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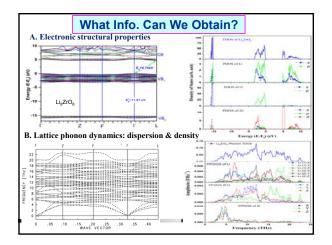


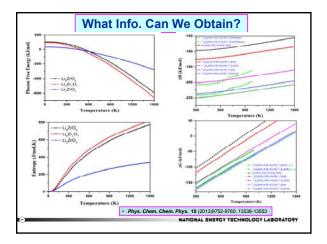


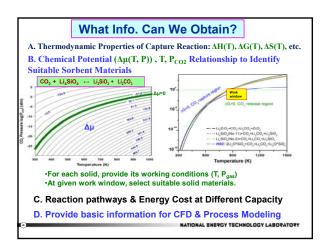


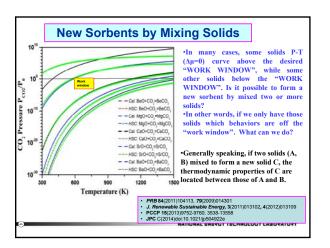


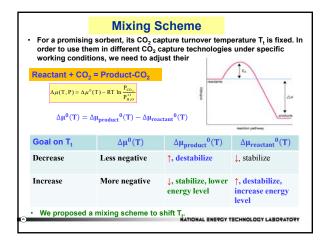


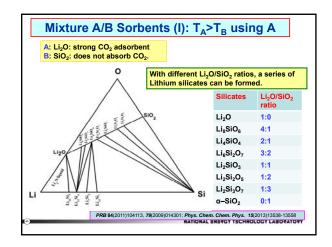


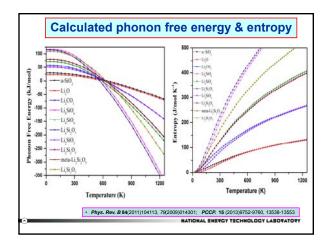


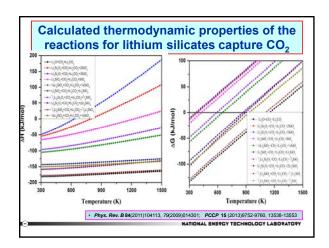


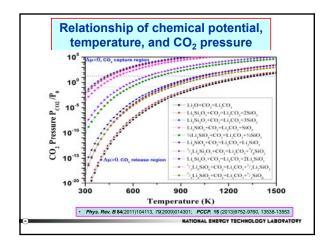


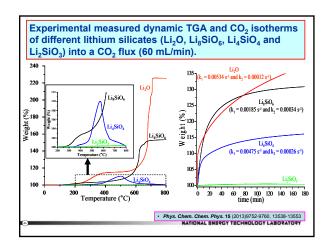




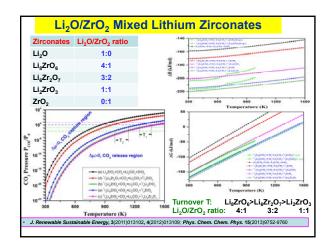


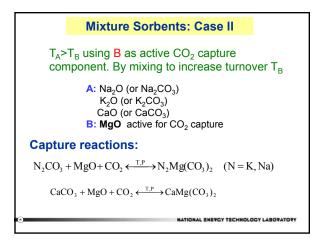


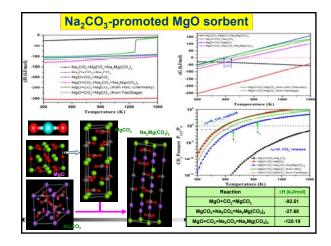


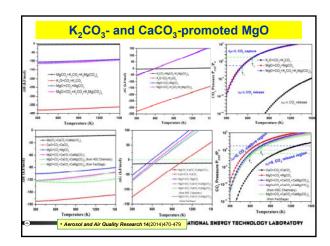


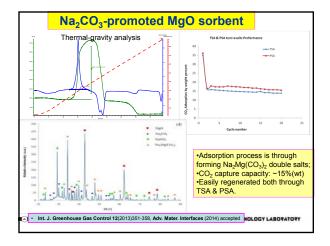




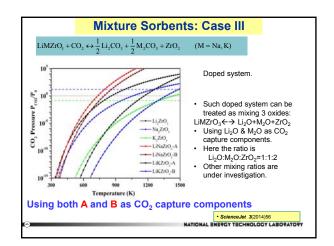


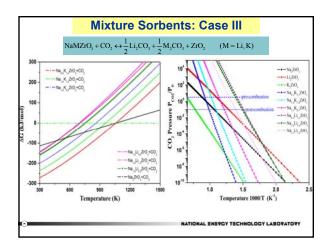






The turnover temperature (T_t) at $P_{CO2} = 1$ CO ₂ at post-combustion (T_1) condition with $P_{CO2} = 10$ bar			
Reactions	T _t (K)	Post-combustion T1(K)	Pre-combustion T ₂ (K)
$MgO + Na_2CO_1 + CO_2 = Na_2Mg(CO_2),$	795	710	915
$MgO + K_2CO_3 + CO_2 = K_2Mg(CO_3)_2$ $MgO + K_2CO_3 + CO_2 = K_2Mg(CO_3)_2$	600	545	695
$MgO + CaCO_1 + CO_2 = CaMg(CO_1),$	660	600	740
	695 ^a , 705 ^b	635 ^a , 630 ^b	785 ^a , 790 ^b
$MgO + CO_2 = MgCO_3$	590	535	685
ingo oot ingoot	575ª, 675b	520 ^a , 605 ^b	655ª, 760b
$CaO + CO_2 = CaCO_3$	1095	975	1245
$Na_2O + CO_2 = Na_2CO_3$	hT°	hT	hT
$K_2O + CO_2 = K_2CO_3$	hT	hT	hT
* Calculated by the HSC Chemistry package b Calculated by the FactSage package c hT means the maximum temperature exceeds our temperatu	re range (1500K).		
mixing Na ₂ O, CaO, K ₂ O into I eases; th process modeling by assur pretical value, the overall perf	ning their	CO ₂ capacities	are half of th



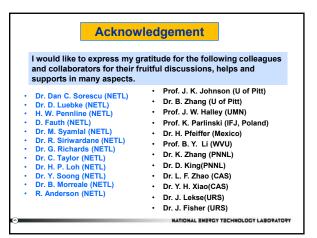


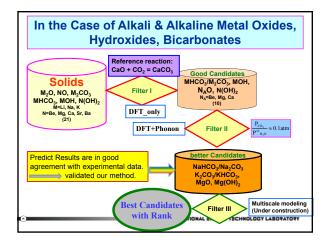
$NaMZrO_3 + CO_2 \leftrightarrow \frac{1}{2}Li_2$	$CO_3 + \frac{1}{2}N$	$I_2CO_3 + Z$	CrO ₂	(M =	= Li, K)			
III. The weight percentage of CO ₂ capture, the ratio and the thermodynamic properties (AH, AG) of the CO ₂ re by solids under the conditions of pre-combustion (H)	capture reactions.	(unit: kJ/mol). T	he turnover	temperatur	es (T1 and T2	the zero-poin) of the react	t energy c ions of CO	hanges D ₂
reaction	absorbing CO ₂ Wt%	Na ₂ O:M ₂ O:ZrO ₂ ratio	AETPT	ΔEP	ΔH (T=300K)	лG (T=300К)	Turnow T ₁	er T (K) T ₂
$\mathrm{Na_2ZrO_3+CO_2 \leftrightarrow Na_2CO_3+ZrO_2^*}$	23.76	1:0:1	-140.862	2.236	-158.327	-114.121	1275	925
$\mathrm{Na}_{i_2}\mathrm{Li}_{i_2}\mathrm{ZrO}_3\text{-}\mathrm{B+CO}_2 \leftrightarrow {}^{1/}_{6}\mathrm{Na}_{3}\mathrm{CO}_{3} + {}^{1/}_{6}\mathrm{Li}_{2}\mathrm{CO}_{3} + \mathrm{ZrO}_{2}$	24.83	$^{3/}e^{3/}e^{1}$	-170.881	4.667	-242.090	-159.144	805	715
$\mathrm{Na}_{i,g}\mathrm{Li}_{i,g}\mathrm{ZrO}_{3}\text{-}\mathrm{B+CO}_{2} \leftrightarrow {}^{1}/_{2}\mathrm{Na}_{3}\mathrm{CO}_{3} + {}^{1}/_{2}\mathrm{Li}_{2}\mathrm{CO}_{3} + \mathrm{ZrO}_{2}$	26.01	%:1/2:1	-157.839	6.480	-228.381	-142.555	745	675
$Na_{i_{2}i_{1}}Li_{i_{2}}ZrO_{3}\text{-}A\text{+}CO_{2} \leftrightarrow {}^{i_{1}}_{i_{1}}Na_{3}CO_{3}\text{+}{}^{i_{2}}_{i_{2}}Li_{2}CO_{3}\text{+}ZrO_{2}$	27.31	56%1	-169.827	9.652	-237.765	-146.230	735	665
$\mathrm{Na}_{1,3}\mathrm{K}_{3,3}\mathrm{ZrO}_3\text{-}\mathrm{A}\text{+}\mathrm{CO}_2 \leftrightarrow {}^{1/}_2\mathrm{Na}_2\mathrm{CO}_3 \text{+}{}^{1/}_2\mathrm{K}_2\mathrm{CO}_3\text{+}\mathrm{ZrO}_2$	22.77	$^{3/}e^{2/}e^{2}$	-210.081	2.486	-281.253	-199.996	915	825
$Na_{1,0}K_{1,0}ZrO_3\cdot B^+CO_2^{1/}_2Na_fCO_3+{}^{1/}_2K_fCO_3+ZrO_2$	21.86	56.5/2:1	-245.436	2.058	-316.736	-236.789	1015	915
$Na_{0.3}K_{1.3}ZrO_3\text{-}B\text{+}CO_2 \leftrightarrow ^{1/}_0Na_2CO_3\text{+}^{1/}_0K_2CO_3\text{+}ZrO_2$	21.02	%2%4:1	-278.147	1.519	-349.077	-272.038	1125	1015
$Li_2ZrO_3{+}CO_2 \leftrightarrow Li_2CO_3{+}ZrO_2 \xrightarrow{h}$	28.75	0:1:1	-146.648	11.311	-158.562	-103.845	1000	780
$K_2ZrO_3{+}CO_2 \leftrightarrow K_2CO_3{+}ZrO_2 \wedge$	20.24	0:1:1	-223.158	5.813	-238.490	-187.884	hT*	1285
Ref 12. Ref 12. ans the temperature is higher than 1500K.								

reaction absorbing CO, W/% ALS ²⁰⁷ (1.900K) AE ²⁰⁷ (1.900K) M (1.900K) AL (1.900K) T, T, T, T, T, T, T, T, T, T, T, T, T, T, T, T, T, <th></th> <th></th> <th></th> <th></th> <th></th> <th>so listed.</th> <th>Turnover</th> <th>T (K)</th>						so listed.	Turnover	T (K)
$\begin{array}{ c c c c c c c c c c c c c c c c c c c$	reaction	absorbing CO ₂ Wt%	ΔE ^{DFT}	ΔE^{2P}	ΔH (T=300K)	ΔG (T=300K)		<u> </u>
$\begin{array}{c ccccccccccccccccccccccccccccccccccc$	Li ₂ ZrO ₂ +CO ₂ ↔Li ₂ CO ₂ +ZrO ₂ *	28.75	-146.648	11.311	-158.562	-103.845	· ·	
$\begin{array}{ c c c c c c c c c c c c c c c c c c c$	LiNaZrO ₃ A+CO ₂ +>%(Li ₂ CO ₃ +Na ₂ CO ₃)+ZrO ₂	26.01	-152.936	7.069	-176.666	-110.892	805	685
$\begin{array}{c c c c c c c c c c c c c c c c c c c $	LiNaZrO ₃ B+CO ₂ ↔%(Li ₂ CO ₃ +Na ₂ CO ₃)+ZrO ₂	26.01	-167.872	6.934	-191.526	-126.477	865	745
$\begin{array}{c c c c c c c c c c c c c c c c c c c $	LiKZrO, A+CO,↔%(Li,CO,+K,CO,)+ZrO,	23.75	.264115	6.006	-287 513	.225.611	1275	1095
K20,-C0,-K00/20,- 2024 -223158 5431 -238.490 -187.884 517 1285 sults: Io be CO2 sorbent the performance of LiNaZrO3 is better than that of Li2ZrO3, but he performance of LiKZrO3 is much worse as its regeneration T is higher. Sy doping with Na into Li2ZrO3, the obtained new solid LiNaZrO3 is a better CO2 sorbent applying to post-combustion capture technology; Sy doping K into Li2ZrO3 to form new solid LiKZrO3, from the thermodynamic point	LiKZrO ₁ B+CO ₃ ↔%(Li ₃ CO ₃ +K ₃ CO ₁)+ZrO ₃			7.080			hT ^b	1285
sults: To be CO_2 sorbent the performance of LiNaZrO ₃ is better than that of Li ₂ ZrO ₃ , but he performance of LiKZrO ₃ is much worse as its regeneration T is higher. By doping with Na into Li ₂ ZrO ₃ , the obtained new solid LiNaZrO ₃ is a better CO ₂ sorbent applying to post-combustion capture technology; By doping K into Li ₂ ZrO ₃ to form new solid LiKZrO ₃ , from the thermodynamic point	Na ₃ ZrO ₁ +CO ₃ ↔Na ₃ CO ₁ +ZrO ₃ *	23.76	-140.862	2.236	-158.327	-114.121	1275	925
To be CO_2 sorbent the performance of LiNaZrO ₃ is better than that of Li ₂ ZrO ₃ , but he performance of LiKZrO ₃ is much worse as its regeneration T is higher. By doping with Na into Li ₂ ZrO ₃ , the obtained new solid LiNaZrO ₃ is a better CO ₂ sorbent applying to post-combustion capture technology; By doping K into Li ₂ ZrO ₃ to form new solid LiKZrO ₃ , from the thermodynamic point	K ₃ ZrO ₁ +CO ₃ ↔K ₃ CO ₁ +ZrO ₃ *	20.24	-223.158	5.813	-238.490	-187.884	hT ^b	1285
mprovements over Li_2ZrO_3 on its overall CO_2 capture performance.	To be CO ₂ sorbent the performance of LiKZrO ₃ By doping with Na into Li ₂ Zr	is much O ₃ , the o mbustion	worse a obtained capture	is its re new s techn	egeneratio olid LiNa ology;	on T is hig ZrO₃ is a b	her. better C	0 ₂

 Our methodology can predict thermodynamic properties of solid materials and their CO₂ capture reactions, which can be used to identify good candidates from vast of material databank. Single solid may not satisfy the industrial operating conditions as a CO₂ sorbent, however, by mixing two or more solids, the new formed solid may satisfy the industrial needs by shifting the turnover T into practical operation range. These results provide guidelines to synthesize 	
conditions as a CO ₂ sorbent, however, by mixing two or more solids, the new formed solid may satisfy the industrial needs by shifting the turnover T into practical operation range.	e
> These results provide guidelines to synthesize	o e
sorbent materials by mixing different solids with different ratio.	







CO_2 captured by carbonates. T ₁ ref	· · · · ·							
Reactions	CO ₂ wt%	ΔE _{DFT} (kJ/mol)	ΔE _{ZP} (kJ/mol)	ΔH (T=300K) (kJ/mol)	ΔG (T=300K) (kJ/mol)	T ₁ (K)	T2 (K)	T _{tr} (K)
K_2CO_3 ·1.5 $H_2O + CO_2$	26.88	-40.47	-0.74	-40.68	-12.82	580 ^b	370 b	445 b
$= 2 KHCO_3 + 0.5 H_2O(g)$						665° 510 ^d	395° 335 ^d	395° 515 ^d
K 60 + 60 + H 0()	31.84	-154.43	18.29	-141.73	-46.28	490 ^b	420 ^b	
$K_2CO_3 + CO_2 + H_2O(g)$ = 2KHCO_3				-142.85ª	-44.72ª	455° 515 ^d	395° 445 ^d	
⁶ Calculated by Chemistry package ¹⁶ ^b when P ₁₀₀₀ = 1 bar ^c when P ₁₀₀₀ = 0.1 bar								
^d when P _{H2O} = 0.1 bar ^d when P _{H2O} = 10 bar								

Summary of the Methodology

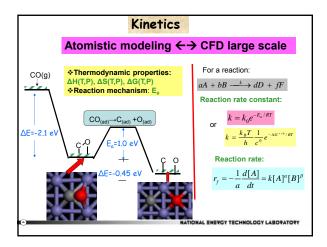
4 Pros:

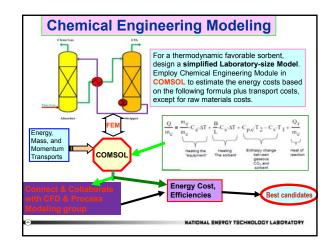
- Performance: the methodology can be used to search for solid > materials with improved CO2 capture performances;
- Reliability: For a given dataset, the promised candidates are always in the final short list and are not screened out;
- Economics: Only promised candidates are needed for experimental tests, which can speed up the searching process and save money;
- Forward predictions:
- Exploring new materials with unknown thermodynamic properties; Provide guidelines for future experimental work.
- Generally, this methodology can be expanded to other classes of solid compounds as well as solutions (ab initio +MD)
- 4 Cons:
 - Need to know the structures of materials with phase transitions; ۶
 - High accurate thermodynamic data (<10 kJ/mol) is not achievable; >
 - High computational demand for complicated systems

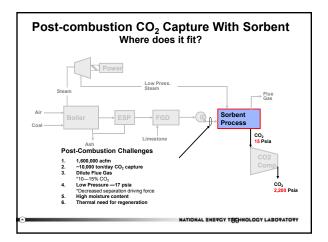
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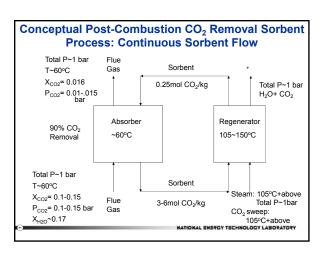
Further Work

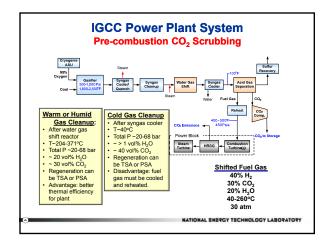
- **4** Using our computational screening methodology continue to explore different classes of oxides, multi-component salts, mixed/substituted/doped solids, etc.
- **4** Exploring the kinetics of the capture process to obtain the activation energy and reaction rates;
- Collaborate with CFD and Process modeling group, establish multi-scale modeling on capture process. (Atomistic-kinetic-CFD-process modeling).
- Based on screening results, experimental measurements will be conducted on those promising candidates;
- 4 Build a database of solid sorbents to satisfy industrial requirements. NATIONAL ENERGY TECHNOLOGY LABORATORY

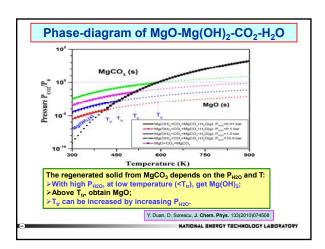


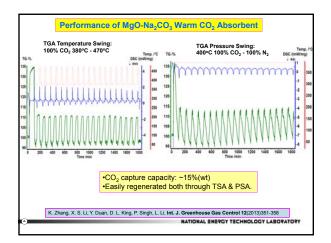


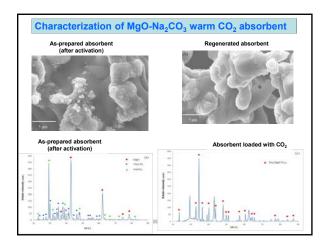








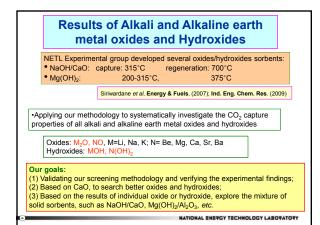


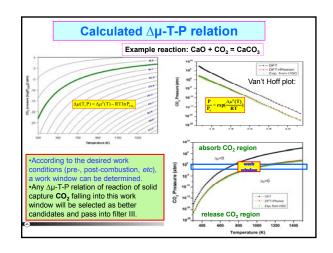


Conclusions

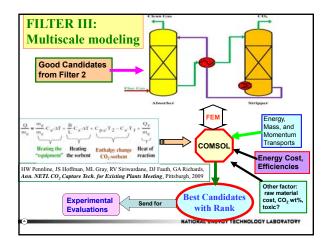
- Our methodology can predict thermodynamic properties of solid materials and their CO₂ capture reactions.
- ➢ By exploring series of lithium silicates with different Li₂O/SiO₂ ratio, we found that with decreasing Li₂O/SiO₂ ratio the corresponding silicate has a lower turnover temperature, and vice versa.
- Compared to pure MgO, the Na₂CO₃, K₂CO₃ and CaCO₃ promoted MgO sorbent has a higher turnover T.
- These results provide guidelines to synthesize sorbent materials by mixing different solids with different ratio.
- Single solid may not satisfy the industrial operating conditions as CO₂ sorbent, however, by mixing two or more solids, the new formed solid may satisfy the industrial needs.
- Using our computational screening methodology continue to explore different classes of oxides, multi-component salts, mixed/substituted/doped solids, *etc.*Exploring the kinetics of the capture process to obtain the activation energy and reaction rates;
 Collaborate with CFD and Process modeling group, establish multi-scale modeling on capture process. (Atomistic-kinetic-CFD-process modeling).
 Based on screening results, experimental measurements will be conducted on those promising candidates;
 Build a database of solid sorbents to satisfy industrial requirements.

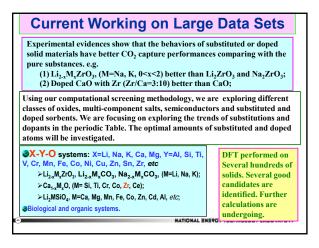
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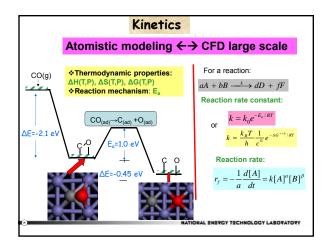


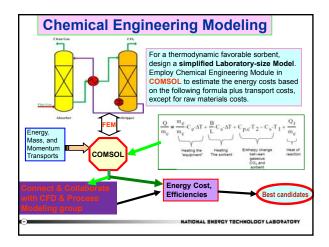


Further Work



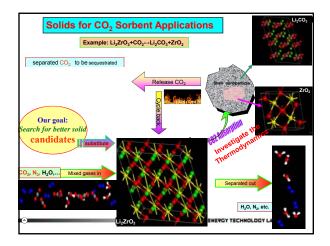


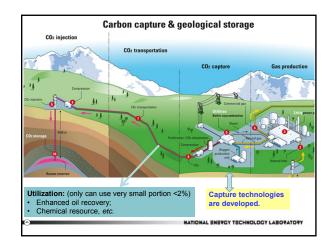


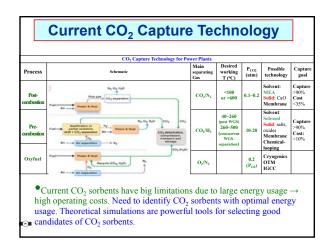


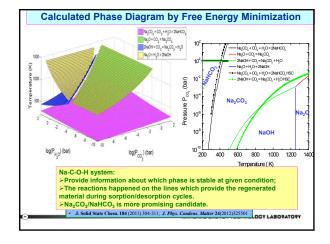
Absorbe nt	Reactions for CO ₂ capture	ΔH°, K J/mol. CO ₂	T,°C for CO ₂ capture	T, °C for CO ₂ release	Capacity, mmol CO ₂ /§
MEA	2MEA + CO ₂ ↔ MEACOO ⁻ + MEAH ⁺	-167	40-65	100-150	8.2
K ₂ CO ₃	$K_2CO_3 + CO_2 + H_2O \leftrightarrow$ 2KHCO ₃	-143	~50	350-400	7.2
Na ₂ CO3	$Na_2CO_3 + CO_2 + H_2O \leftrightarrow$ 2NaHCO ₃	-136	60-70	100-200	9.4
Li ₂ SiO ₃	$Li_4SiO_4 + CO_2 \leftrightarrow Li_2CO_3 + Li_2SiO_3$	-142	600	850	8.3
CaO	$CaO + CO, \leftrightarrow CaCO,$	-178	600-700	800-900	17.9

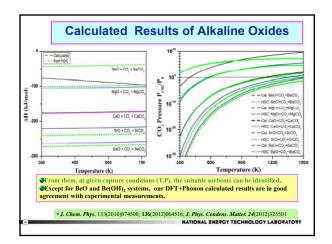
Materials	BET surface area	TGA Performance CO ₂ wt%			
	m²/g	Initial	10 Cycles.		
Nano MgO	702	3.1	NA		
5C synthesized	19.9	49.7	2.0		
17C synthesized	16.8	50.6	5.2		
24C synthesized	15.6	40.1	14.7		
30C synthesized	9.3	42.5	15.1		
37C synthesized	2.9	24.0	12.2		
Slowly mix Na	table MgO-Na ₂ C($_2$ CO ₃ solution and	•			
	₂ CO ₃ solution and	•			
I. Slowly mix Na ~30 °C. White	₂ CO ₃ solution and	Mg(NO ₃) ₂ solu	tion at		
. Slowly mix Na ~30 °C. White 2. Settle the slur	₂ CO ₃ solution and slurry forms.	Mg(NO ₃) ₂ solu	tion at		
I. Slowly mix Na ~30 °C. White 2. Settle the slur 3. Filtration unde	₂ CO ₃ solution and slurry forms.	Mg(NO ₃) ₂ solu	tion at		
. Slowly mix Na ~30 °C. White . Settle the slur . Filtration unde . Dry the solid p	₂ CO ₃ solution and slurry forms. ry overnight at roo er vacuum.	Mg(NO₃)₂ solu m temperature C.	tion at		

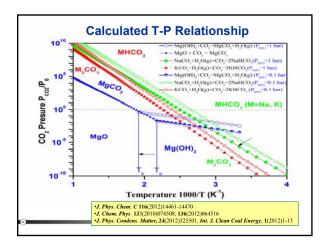


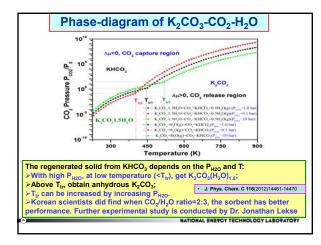


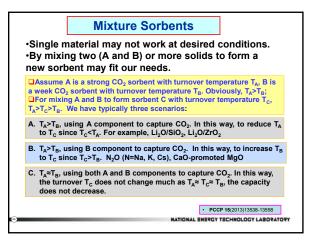












Summary	/ of t	he Ca	aptu	re R	eacti	ons	
reaction	Wt%	ΔE_{DFT} (kJ/mol)	ΔE_{ZP} (kJ/mol)	$\Delta \mathbf{H}$ (kJ/mol)	∆G (kJ/mol)	Т ₁ (К)	Т2 (К)
Li ₂ O +CO ₂ ↔ Li ₂ CO ₃	147.28	-203.96	4.52	-180.89	-133.42	1375	1025
${}^{l}_{3}Li_{8}SiO_{6}{+}CO_{2}{\leftrightarrow}Li_{2}CO_{3}{+}^{l}_{3}Li_{2}SiO_{3}$	73.51	-181.97	5.16	-177.37	-129.11	1325	985
$^{1}\!\!4\mathrm{Li}_8\mathrm{SiO}_6\mathrm{+CO}_2\mathrm{\leftrightarrow}\mathrm{Li}_2\mathrm{CO}_3\mathrm{+}^{1}\!\!4\mathrm{SiO}_2$	98.01	-154.99	5.42	-150.13	-102.04	1115	835
$\rm Li_4SiO_4+CO_2 {\leftrightarrow} \rm Li_2CO_3+\rm Li_2SiO_3$	36.72	-148.78	5.97	-143.55	-94.05	1025	775
$^{1/2}Li_4SiO_4+CO_2\leftrightarrow Li_2CO_3+^{1/2}SiO_2$	73.44	-111.42	6.07	-104.36	-57.44	765	585
$Li_6Si_2O_7 {+} CO_2 \leftrightarrow Li_2CO_3 {+} 2Li_2SiO_3$	20.98	-169.26	13.09	-158.09	-104.38	1045	805
${}^{l_3}Li_6Si_2O_7{+}CO_2 \leftrightarrow Li_2CO_3{+}^{l_3}SiO_2$	62.93	-105.79	8.48	-96.08	-48.68	695	525
Li ₂ SiO ₃ +CO ₂ ↔Li ₂ CO ₃ +SiO ₂	48.92	-74.06	6.18	-65.08	-20.83	505	385
$\rm Li_2Si_2O_5{+}CO_2 \leftrightarrow \rm Li_2CO_3{+}2SiO_2$	29.33	-66.75	5.90	-54.06	-13.87	455	345
$Li_2Si_3O_7 \!$	20.94	-63.83	5.52	-48.60	-11.26	435	335
Phys. Rev. B 84	(2011)1041	13, 79(2009)()14301;; <i> </i>	PCCP 15 (2	013)9752-97	60, 13538	-13553
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he reaction	mic properties (ΔΗ, ΔG) of the C is of CO ₂ capture by solids un r) are also listed.	by solids under the conditions of pre-co		istion (P _{coz} =2		post-con		
atio	reaction	absorbing CO ₂ Wt%	ΔE^{DFT}	ΔE^{ZP}	∆H (T=300K)	ΔG (T=300K)	Т,	T2
i ₈ ZrO ₆	¹ / ₄ Li ₈ ZrO ₆ +CO ₂ ↔Li ₂ CO ₃ + ¹ / ₄ ZrO ₃	72.50	-185.495	5.681	-200.762 -203.393 ^b	-151.604 -153.432 ^b	1460	1090
:1	%Li ₈ ZrO ₆ +CO ₂ ↔Li ₂ CO ₃ + %Li ₂ ZrO ₃	54.38	-206.945	3.804	-215.156	-167.851	>1500	1220
	² / ₅ Li ₈ ZrO ₆ +CO ₂ ↔Li ₂ CO ₃ + ¹ / ₄ Li ₆ Zr ₂ O ₇	45.31	-203.751	3.914	-220.043	-172.551	>1500	1250
i ₆ Zr ₂ O ₇ :2	1 Li ₆ Zr ₂ O ₇ +CO ₂ \leftrightarrow Li ₂ CO ₃ + 3 ZrO ₂ ^a	39.28	-155.942	8.624	-169.500	-117.564	1140	880
.i ₂ ZrO ₃ :1	$Li_2ZrO_3{+}CO_2{\leftrightarrow}Li_2CO_3{+}ZrO_2$	28.75	-146.648	11.311	-158.562	-103.845	1000	780
Turnov Li ₂ O/Z	ver T: Li ₈ ZrO ₆ > rO ₂ ratio: 4:1	Li ₆ Zr ₂ O ₇ 3:2	>Li ₂ ZrO 1:1	3				
	Question	: Can Li _s	ZrO ₆ &	Li ₆ Zr ₂	O ₇ be fu	lly reger	nerate	d?